

Microstructure and Mechanical properties of Multipass Friction Stir Processed Aluminum Silicon Carbide Metal Matrix

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Abstract: *In this study, SiC particles were incorporated by using Friction Stir Processing (FSP) into the 6351 aluminium alloy to form particulate composite materials. Samples were subjected to constant rotational and traverse speeds of the FSP tool with and without SiC reinforcements. Microstructural observations were carried out by employing optical microscopy of the modified surfaces. Mechanical properties were evaluated by tensile test on UTM. For the 100% overlapping, No. of passes caused grain modification in the processed zone. The tensile test results indicate an improvement of strength and Microhardness for Single Pass FSP samples and a reduction in the strength of FSP samples with increasing the No. of passes due to the dissolution of the hardening precipitates.*

1. Introduction

Al alloys of the 6000 series are known to have good formability, corrosion resistance, weldability, and high strength-to-weight ratio. They have constituted the highest volume of Al extruded product and been widely used for automobile and aerospace industries, architectural applications, bicycle frames, transportation equipment, bridge railings and welded structures. The major alloying elements in the heat-treatable 6000 series are silicon and magnesium. Both elements are required for precipitation strengthening, which is commonly acquired by solutionizing and artificial aging. Strengthening can be enhanced further by refining the grain size to a few micrometers and lower. This can be achieved by Friction Stir Processing (FSP), which is a solid state processing technique developed from the principles of Friction Stir Welding (FSW) and used for microstructural modification by grain reinforcement. In FSP, a non-consumable rotating tool consisting of pin and shoulder is inserted into a plate of metal and traversed along a line to process the region of interest. Friction between the tool shoulder and the workpiece results in localizing heating that softens and plasticizes the processed zone. The stirring action of the rotating pin causes intense plastic deformation of the locally heated material. The combination of the plastic deformation, mixing and thermal exposure results in a modified microstructure in the stirring zone (SZ), which is commonly characterized by fine and equiaxed grain structure with predominate high-angle boundaries. The modification is generally attributed to dynamic recrystallization and break-up of constituent particles. Next to the SZ, a narrow transition region known as thermomechanically affected zone (TMAZ) is formed, followed by the heat affected zone (HAZ), and finally the unaffected base metal (BM).

FSP was developed by Mishra et al. [1] microstructural modification and fabrication of metal matrix composites are based on the basic principles of the FSW. Several

FSW and FSP works have been devoted to study the microstructural and mechanical properties of 6000 series Al alloys. General remarks indicate that frictional heat and intensive plastic deformation during FSW cause dissolution and/or coarsening of strengthening precipitates. Woo et al. [2] investigated the influence of stirring pin and pressing tool shoulder on the microstructural development in FSP of 6061 Al. It was found that microstructural softening occurred due to frictional heating by the tool shoulder. The effect of multiple passes on microstructural development during FSP has been investigated on different aluminium alloys [3,4]. Johannes and Mishra [5] demonstrated the effectiveness of multiple passes in creating large areas of superplastic material of 7075 Al with insignificant microstructural difference, having grain size in the range of 3.6-5.4. Ma et al. [6] also showed similar grain refinement in the microstructure of 7075 Al produced by multi-pass FSP and indicated that two-pass FSP samples of 50% overlap have greater superplasticity than single-pass FSP samples. A recent study by El Rayes and El Danaf [7] on multi-pass FSP of 6082 Al indicated that the increase in the number of passes from one to three with 100% overlapping resulted in large SZ size but more dissolution of the hardening phase. This was found to soften the SZ and reduce its strength.

The influence of the overlapping (OL) percentages of multi-pass FSP on the microstructural and mechanical characteristics of the alloy surface has received less attention. The intersection between consecutive passes creates an overlapped region between the processed zone and thus further alters the microstructures. Also, there is a possibility of additional dissolution and/or coarsening of strengthening precipitates of Al 6351 at the overlapped region due to thermal cycle of subsequent pass. This can progress to further weakening of the mechanical properties of the processed zone. Therefore, the aim of the study is to investigate the effect of the OL percentages produced by multi-pass FSP on the microstructure and mechanical properties of Al 6351. By running Multi-pass FSP with one overlapping percentage it is also possible to examine the comparative analysis of single-pass and multi-pass FSP of Al 6351.

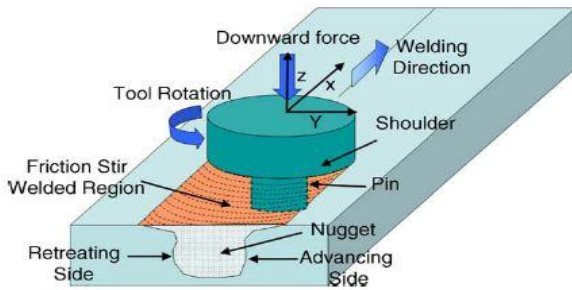
The objective of this paper was to investigate the possibility of incorporation of reinforcement particles into surface layer of commercially Al 6351 to form metal matrix composites by means of FSP technique. Also, the influence of no. of passes on the mechanical properties were experimentally investigated.

2. Experimental procedure:

Four millimetre thick plates of Aluminium alloy 6351 plates were used in this study. The dimensions of the FSP plates were 100mm x 50mm. The FSP

tool material is made of HSS steel ; it has a shoulder of 16mm dia and 2mm long pin with a (2*3)mm taper dia. FSP was conducted using a vertical milling machine with constant tool rotating rate 1000rpm and travel speed of 20mm per minute. It should be note that WD,TD and ND denotes the processing ,transfers and normal directions of the plate respectively.Single pass & multi pass FSP were carried out with and with out SiC. The surface plates was cleaned with grinding paper before processing. The average size of the SiC particles was about 10µm. In a small group cut on the work piece and filled with SiC particles then FSP was carried out.In the case of multi pass processing, three passes were made with 100% overlapping .

To asses microstructure refinement in the SZ, metallographic samples were prepared for examination by using optical microscopy. The metallographic samples were cut from the cross section of FSP plates, then ground and polished using standard metallo graphy techniques.The optical microscopy(OM) studies were carried out the samples were chemically etched at ambient temperature for about 6 min with kellers reagent to distinguish the grein boundaries, identify precipitates and difference in composition. Further examination in SZ was also made by X-ray diffraction. The mechanical properties of FSP samples were measured at ambient temperature by tensile testing, TD tensile specimens of 4-mm thick were cut from the FSP plates a guage length of 50mm was used to ensure that the guage section is entirely made from the processed zone. For each processing condition, the tensile specimens were made and tested by using the computerized UTM. The tensile strength was then computed and compared to the tensile strength of the base metal.



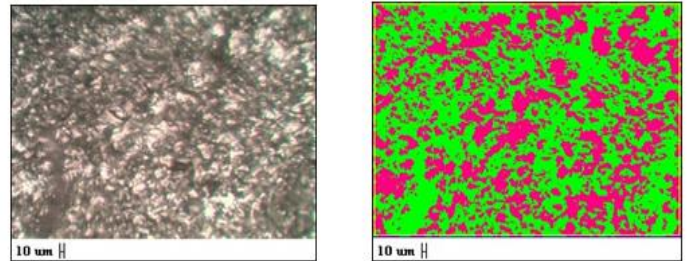
3. Result and Discussion:

3.1. Microstructure:

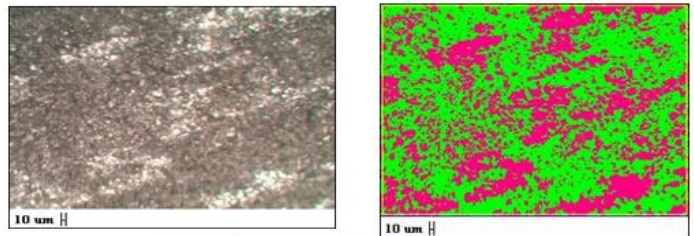
After the Aluminium 6351 Alloy-SiC composite was fabricated by Friction Stir Processing, microstructure at cross-section of stir zone was observed using optical microscope. The prcessing parameters are 1000rpm rotational speed of tool and 20mm/min transverse speed. The average size of the SiC particles is 10µm. The microstructure of base materiaal 6351 Alloy after conducting friction stir processing in single, two and three passes are shown in fig. Respectively. As it is already observed from the literature, that rotational speed and travelling speed variations varies the microstructure significantly, In the presant study all the experiments are conducted at 1000rpm rotational speed and

20mm/min traverse speed. From fig(a) for single pass it is observed that around 43.6% by volume Silicon flakes are present in α-Al matrix. As the no.of passes increased, with 100% overlap a refinement of Silicon flakes is observed and volume % of silicon increased to 44.81%. In the third pass, the Silicon flakes are futher refined and vollume % of Silicon is found to be 42.8%. It is noticed that the grain size is reduced for 100% overlapping specimens compared to specimens with single pass(without overlapping).

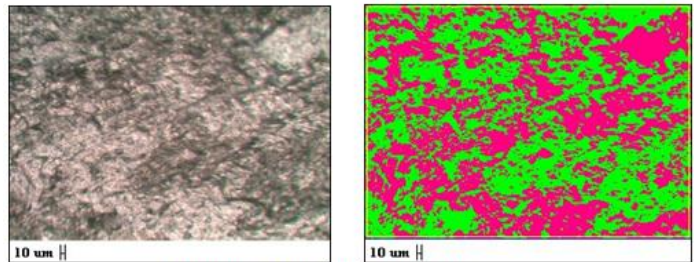
Fig. Shows the micrographs of Al 6351+ SiC composite specimens in the stirring zone. The addition of SiC particles into Al 6351 matrix using the Al 6351 matrix. With multipass friction stir processing with 100% overlap resulted in more heat input and so grain growth increased and SiC is distributed non-homogeneously in Al matrix.



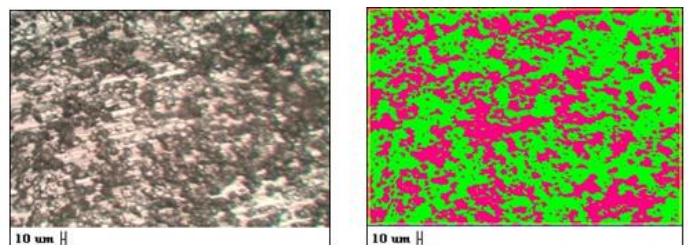
Fig(1) Microstructure of Al6351 Single pass



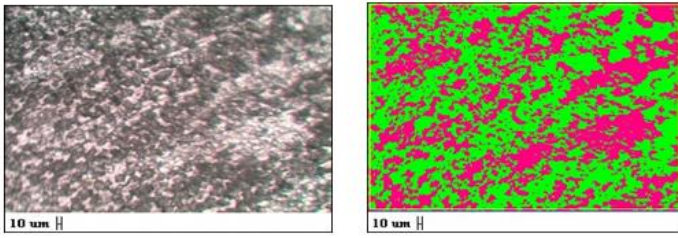
Fig(2) Microstructure of Al6351 Two pass



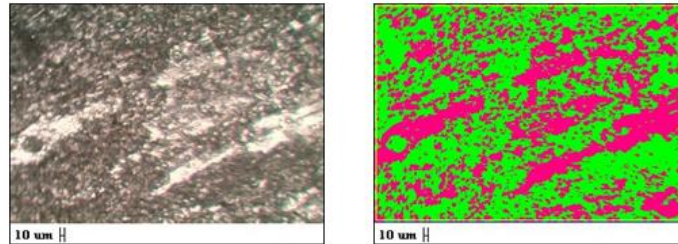
Fig(3) Microstructure of Al6351 Three pass



Fig(4) Microstructure of Al6351+SiC single pass



Fig(5) Microstructure of Al6351+SiC double pass



Fig(6) Microstructure of Al6351+SiC three pass

3.2. Mechanical properties:

3.2.1. Tensile Properties:

Table shows the results of tensile properties obtained after experimentation. It is observed that the UTS, yield strength increases with single pass friction stir processing for base material Al 6351 alloy. However, with multipass using 100% overlapping the same properties showed a decreasing trend. This may be due to refinement of grain size with single pass FSP and softening of grains due to excessive heat input during multipass leading to decrease in tensile properties for multipass FSP specimens.

The addition of SiC reinforcement in Al 6351 base alloy resulted in improvement of tensile properties like UTS and yield strength. This is due to homogeneous distribution of SiC particles in Al base alloy and good bonding between SiC particles and base Al. The multi-pass FSP with SiC reinforcement reduced the tensile properties of specimens, probably due to grain softening and non-uniform distribution of SiC in the base alloy matrix.

Tensile properties of Aluminium Alloy 6351-SiC surface composite				
Material	Type of specimen	UTS(mpa)	Yield strength(mpa)	% Elongation
Al 6351 alloy (base metal)	Single pass	239.16	232.4	9.2
	Two pass	151.24	147.55	22.6
	Three pass	142.87	139.66	17.6
Al 6351 alloy +SiC	Single pass	158.49	137	15.8
	Two pass	146.25	127.97	14
	Three pass	120.87	104.27	6.6

Table 1: Tensile properties of Friction Stir processed specimens

3.2.2. Hardness:

The results of the vickers microhardness values for all the test samples are shown in table. The hardness of Al 6351 base alloy with single pass has increased compared to Al

6351 base alloy without FSP. For the multi-pass test specimens the hardness is found to be decreased due to grain softening caused by excessive heat input and non-uniform distribution of silicon. For Al 6351-SiC composite with single pass, the hardness is found to increase compared to Al 6351 base Alloy with single pass, this is due to the inclusion of hard SiC particles into the Al matrix. However the effect of multi-pass FSP on hardness of both Al6351 base alloy and Al 6351-SiC composite is found to be decreasing due to heat input and grain softening during multipass FSP.

Material	No. of passes	Center	Left to Right			
Al 6351	1	135.9	136.6	135	138.9	136
	2	120.6	121	120.3	119.8	120.9
	3	105.5	104.9	104	105.3	103.8
Al6351+SiC	1	138.7	138.2	137.4	137.9	138.1
	2	126.2	126.8	126	126.3	125.8
	3	119	118.7	119.1	119.4	118.9

Table 2 : Hardness values

4. Conclusions:

1. It has been demonstrated that FSP was an appropriate method to modify the microstructure and mechanical properties of 6351 Al-alloy.
2. FSP resulted in fine and equiaxed grains in the stirring zone and overlapping area.
3. Increasing the percentage of overlapping resulted in limited change in the grain size, but provided more subgrain formation.
4. A reduction in the strength and microhardness of multi-pass FSP samples occurred due to the precipitate dissolution and the limited reprecipitation by the thermal cycle of FSP.
5. The highest microhardness of Al 6351/SiCp composite can be attributed to the presence of reinforcement particles.
6. With further research efforts, FSP could be conducted for mechanical behaviour of these composites, like fatigue, wear and creep, response etc...

References

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